

Photonic Metamaterial with Sub-Wavelength Electrode Pattern

GUILLAUME CROES^{1,2,4}, RENAUD PUYBARET¹, JANUSZ BOGDANOWICZ¹, UMBERTO CELANO^{1,3}, ROBERT GEHLHAAR¹ AND JAN GENOE^{1,2}

¹ imec, Sensors and Actuators Technology (SAT) Dept., Kapeldreef 75, 3001 Leuven, Belgium

² Electrical Engineering Dept. (ESAT), Katholieke Universiteit Leuven, 3001 Leuven, Belgium

³ School of Electrical, Computer and Energy Engineering, Arizona State University, Tempe, AZ, 85287 USA

⁴Guillaume.Croes@imec.be

<https://orcid.org/0000-0001-6168-9794>

Abstract: The next generation of tunable photonics require highly conductive and light inert interconnects that enable fast switching of phase, amplitude and polarization modulators without reducing their efficiency. As such, metallic electrodes should be avoided as they introduce significant parasitic losses. Transparent conductive oxides, on the other hand, offer reduced absorption due to their high bandgap and good conductivity due to their relatively high carrier concentration. Here, we present a metamaterial that enables electrodes to be in contact with the light active part of optoelectronic devices without the accompanying metallic losses and scattering. To this end, we use transparent conductive oxides and refractive index matched dielectrics as the metamaterial constituents. We present the metamaterial construction together with various characterization techniques that confirm the desired optical and electrical properties.

© 2023 Optica Publishing Group

1. Introduction

Large area photonic devices as metasurfaces, optical phased arrays and spatial light modulators (SLM) have shown excellent control over incident light, whether waveguide fed or plane wave illuminated. Their large degree of freedom is unique and enabled them to bring new perspectives to applications such as wavefront shaping and holography. [1] They achieve these feats due to careful design of the modulators that influence phase, amplitude and polarization. Evidently, a vast array of technologies including thermo-optics [2], microelectromechanical systems [3, 4], liquid crystals [5, 6] and phase change materials [7, 8] are being explored to enable fast and reliable modulator switching. Some of these state-of-the-art devices, such as metasurfaces, cannot be switched, while others are currently controlled optically. [9, 10] In either case, electrical modulation is preferred for more robust switching. However, even when devices are electrically tunable, doing this over a large area remains a challenge due to the combination of electrode losses and desired modulator size. This is due to the link between the attainable field of view and the modulator or pixel size, which states that accurate control spanning over a 180° viewing angle requires subwavelength modulation. Hence, factoring in the refractive index of the modulator, light modulation needs to occur at scale of 100nm or less. Consequently, a similar size restriction exists for the accompanying driving electronics needed to individually address each modulator. Due to the shear number ($> 1E7/mm^2$) of modulators that need switching, the most common approach would be to create metallic interconnects which introduce significant parasitic losses into the devices, leading to reduced efficiency.

It is well known that metallic electrodes can impede the operation of photonic, optoelectronic and plasmonic devices such as LEDs, liquid crystal cells and waveguide modulators due to their strong light matter interaction. [11, 12] This behaviour stems from their high carrier concentration,

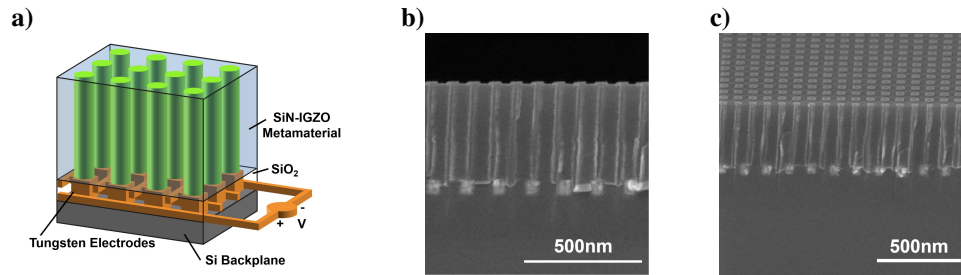


Fig. 1. a) Schematic of the metamaterial. b) Cross section SEM image. c) Angled top down SEM image.

46 which leads to a high conductivity but also to a high absorption due to the unbound nature of
 47 conduction electrons. The lowered performance can be avoided by taking special care with the
 48 device design. In practice, this often results in physically separating the metal from the active
 49 region which is not an ideal solution for high modulator densities. Alternatively, a wide range of
 50 materials are being considered to replace metals as electrode including transparent conducting
 51 oxides (TCOs) and diluted metals. [13–16]

52

53 Transparent conductive oxides form a group of materials that are easily deposited, CMOS
 54 compatible, highly optically tunable [17] and have carrier densities similar to highly doped
 55 semiconductors [18], hence well below those of metals. On top of that, they are semi-transparent
 56 to transparent due to their high bandgap and have been extensively used in thin film devices
 57 as LEDs and touch screens. Nowadays, these material are becoming more widespread, finding
 58 uses in metasurfaces [19–22], epsilon near zero materials [23–25] and electro-optical modula-
 59 tors [26–28]. TCOs are often used in tandem with metals due to their non-negligible ohmic
 60 losses. Hence, they typically form a bridge between the light active region of the device and the
 61 peripheral electrical circuitry. However, in periodic structures, such as tunable metasurfaces and
 62 spatial light modulators, TCOs are typically not index matched with their surrounding materials.
 63 This again leads to undesired scattering of incident light, depending on the grating formed by the
 64 periodic TCO.

65

66 Here, we present a metamaterial that employs a TCO together with an index matched dielectric
 67 such that electric bias can be applied directly to the active light region. Additionally, this
 68 metamaterial can be used as cladding layer in integrated photonics since the material optically
 69 behaves isotropic in the chosen light range. The employed electrode scale allows sub-wavelength
 70 and individual electrode modulation in the visible range resulting in a 180° viewing angle without
 71 any ghost images or scattering artefacts. Consequently, the created metamaterial enables the
 72 creation of high-density tunable modulators for transmission, reflection or waveguide based
 73 applications such as a holographic display. We showcase the required fabrication techniques
 74 and give insight in the optical and electrical properties by ellipsometry, scatterometry, Rigorous
 75 Coupled Wave Analysis (RCWA) and Conductive Atomic Force Microscopy (C-AFM).

76

77 2. Metamaterial Design and Fabrication

78 A metamaterial was designed to provide the electrical bias required for a waveguide based spatial
 79 light modulator. To that end, three criteria where kept in mind. Namely: (1) steering over a 180°
 80 range is enabled, (2) no light scattering due to a refractive index difference between the electrode
 81 and dielectric material, and (3) no or limited influence from the remaining metallic components.

82

83

Firstly, ensuring a device can steer over the full 180° range requires a modulation at half the size of the desired wavelength. (Diffraction Grating) However, since the envisioned device is waveguide-based, the required modulation is linked to the internal wavelength. Hence, it is the effective refractive index of the guided mode of the envisioned SLM that determines the pitch of these sub-wavelength electrodes. A well confined mode calls for a waveguide refractive index that is higher than the cladding materials. For common strong electro-optic materials, such as Lithium Niobate ($LiNbO_3$) and Barium Titanate ($BaTiO_3$) an electrode pitch of 90nm practically covers the entire visible spectrum. [29–31]

91

92

Secondly, it is the optical properties of the constituent materials of the metamaterial that determine whether Bragg scattering can occur. Figure 1 a) shows a schematic of the envisioned metamaterial. Several TCO's were considered, including Indium Tin Oxide (ITO), Aluminum doped Zinc Oxide (Al:ZnO) and Indium Gallium Zinc Oxide (IGZO). Since atomic layer deposition (ALD) was needed for uniform filling, only ITO and IGZO were at hand in our fab. IGZO was preferred over ITO since it is known to have a lower intrinsic carrier concentration which suits the intended application, as it does not require high switching speeds (video rate) while simultaneously lowering parasitic absorption. For other applications, such as optical I/O, ITO can result in faster switching at the cost of slightly higher optical losses. Silicon Nitride (SiN) was paired to IGZO as dielectric since it is semiconductor fab compatible, easily deposited and tunable over the same refractive index range. We used an IGZO recipe which has proven to yield good conductivity, to which we index matched SiN to ensure the desired electrical properties. [32, 33] This is beneficial since at wavelengths where both refractive indices match, their combination will behave isotropic. On top of that, even if a small refractive index difference exists, the small electrode pitch leads to reduced Bragg scattering since only higher order and thus weaker scattering can take place.

107

108

Finally, to limit undesired absorption from metallic backplane components, the metamaterial is preferred as thick as possible to separate the waveguide physically from any metals. However, for ease of manufacturing the aspect ratio (AR) (depth/width) of the electrode pillars cannot be excessively large. The metamaterial was thus made 500nm thick, resulting in an easily reproducible metamaterial with AR close to 11.

113

114

In practice, the metamaterial waveguide cladding layer is created as follows. First, a TiN / Tungsten back contact on SiOx with critical dimension 50nm is created using a damascene process : 300nm SiOx deposition on Si substrate, lithography 193nm immersion with BARC and positive-tone resist, 100nm etch of SiOx, 10nm TiN ALD, 200nm W fill, CMP for planarization. Afterwards, the metamaterial is created by depositing a thick SiN layer (500nm) which is then subjected to a high AR etch to open 45nm sized holes with 90nm pitch. Atomic layer deposition (ALD) is subsequently used to fill the SiN holes. Due to ALD's structure filling capabilities a conformal IGZO fill can be achieved along the whole pillar as shown in Figure 1 b) - c). Chemical-Mechanical Polishing (CMP) is used to remove the excess IGZO from the top and planarize the metamaterial. During planarization, a small amount of the metamaterial is consumed, resulting in a layer thickness around 420nm. This entire process is done in imec's CMOS 300mm fab. A post-process anneal in an O_2 rich environment at $250^\circ C$ for an hour enables the IGZO conductivity through the creation of oxygen vacancies. [34]

126

127

128 **3. Results and Discussion**

129 **3.1. Optical Characterization**

130 Optical characterization was performed on planar layers (Ellipsometry) and on the finished
131 metamaterial (Scatterometry and RCWA) to gain insight their refractive indices and structural
132 parameters respectively.
133

134 **3.1.1. Ellipsometry**

135 The optical properties of the constituent materials are shown in Figure 2 a) - b). Ellipsometric
136 measurements (Woollam RC2) were performed on planar layers after which the data was fitted
137 using a Tauc-Lorentz oscillator for SiN and a combination of Tauc-Lorentz and Drude oscillators
138 for IGZO and the metamaterial. Here, the Tauc-Lorentz oscillator represents the amorphous
139 nature of both SiN and IGZO, and the Drude oscillator is employed to fit IGZO's metallic infrared
140 absorption. [35, 36].
141

142 To limit the present degrees of freedom, as both IGZO and SiN are highly tunable materials
143 depending on their deposition parameters, a previously optimized recipe for high conductivity
144 IGZO was used. [32, 33] The fitting parameters for IGZO can be found in Table 1. Its resistivity,
145 $0.003681\Omega cm$, extracted from the Drude oscillator is similar to the resistivity of a weakly
146 conducting semiconductor. This approach ensures the desired electrical properties and only
147 requires SiN to be varied by altering its deposition parameters, influencing its stoichiometry and
148 density, until its real refractive index matches well in the visible (440nm-640nm) as indicated in
149 Figure 2 a) and b). Thus limiting the number of required blanket wafers. The total measured SiN
150 range (light blue) as well as the closest matching SiN wafer (black) is shown in Figure 2 a) and
151 b). Its fitting parameters are shown in Table 2. All ellipsometry measurements, including the full
152 SiN range can be found in Data File 1.
153

Table 1. SiN fitting parameter for the employed Tauc-Lorentz oscillator.

SiN Tauc Lorentz Parameter	Amp (eV)	Eo (eV)	Br (eV)	Eg (eV)
Value	57.5564	7.422	5.951	2.654

Table 2. IGZO fitting parameters for a Tauc-Lorentz and Drude oscillator model.

IGZO Tauc Lorentz Parameter	Amp (eV)	Eo (eV)	Br (eV)	Eg (eV)
Value	118.5894	3.540	11.140	3.163
IGZO Drude Parameter	Resistivity (Ωcm)		Scattering time (fs)	
Value	0.003681		2.096	

154 The completed metamaterial was initially fitted with the 3 oscillators expected for a blend of
155 both materials (2 Tauc-Lorentz and 1 Drude) and constrained to a thickness of 420nm measured by
156 SEM. Unfortunately, this model does not result in a good fit as one of the Tauc-Lorentz oscillators
157 yields unrealistic values. The best possible fit, shown in Table 1 of the Supplemental Document,
158 was achieved by using only one Tauc-Lorentz oscillator, for which the refractive indices are shown

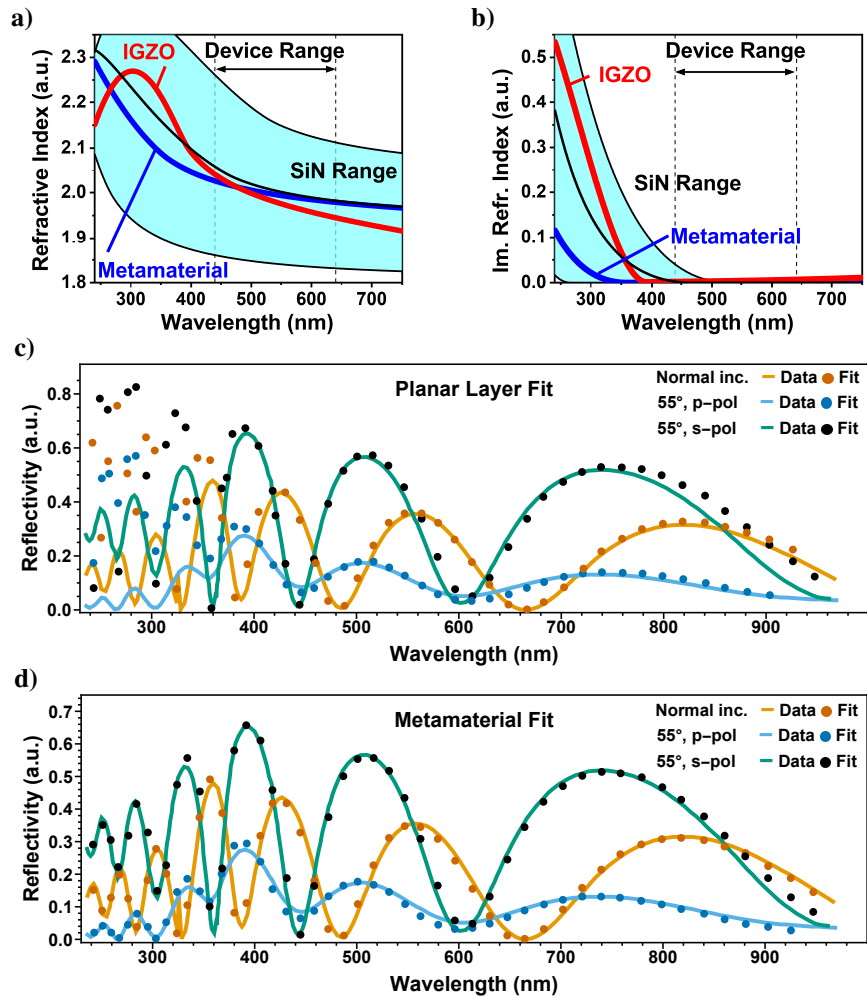


Fig. 2. a) Refractive index (n) data of the metamaterial and its constituent materials measured by ellipsometry. b) Refractive index (k) data of the metamaterial and its constituent materials measured by ellipsometry. c) Scatterometry measurement fitted with a planar layer model. d) Scatterometry measurement fitted with a model having embedded pillar electrodes.

159 in Figure 2 a) and b). That said, even for this best case scenario, the metamaterial fit remains
 160 rather poor and yields an imaginary refractive index that does not align with its constituents.
 161 The fit worsens when the thickness is included as fitting parameter. Here, the model gives a
 162 significantly lower refractive index (both real and imaginary), paired with a larger thickness of
 163 460nm. (See Supplemental Document) This is a rather unrealistic result, indicating that the
 164 inclusion of IGZO inside a SiN matrix can no longer be fitted with an isotropic model. It should,
 165 however, not be a surprise that an anisotropic metamaterial can not be fitted with an isotropic
 166 model. In fact, only at the wavelength where the 2 materials are equal an isotropic model
 167 succeeds. Evidently, a more advanced optical model is needed to extract correct parameters for
 168 which we look towards scatterometry and Mueller matrix measurements.

170 3.1.2. Scatterometry

171 Scatterometry measurements (Nova T600 MMSR) and fits (Nova Mars) were performed on the
 172 finished metamaterial, shown in Figure 2 c) - d). The earlier mentioned refractive indices were
 173 used as input to extract the parameters of the embedded IGZO electrode in the SiN matrix.

174
 175 Figure 2 c) shows the measured data (full line) and a structural fit (points) that represents
 176 the metamaterial by a single SiN layer without embedded electrodes. The fit indicates that this
 177 unpatterned layer should be 411nm thick, which aligns well with the SEM data. For all wave-
 178 lengths above 400nm, the model corresponds well with the measured data. At wavelengths
 179 below 400nm, due to the growing difference in SiN and IGZO refractive indexes, we are not
 180 expecting to obtain a fit in that spectral range. Figure 2 d) shows the same measurement data,
 181 now fitted with a model that assumes a SiN layer with embedded IGZO electrodes. The fit
 182 indicates a metamaterial thickness, pillar height and pillar width of 408nm, 423nm and 74nm
 183 respectively. For this second model, the complete wavelength range was fitted well, showcasing
 184 that the created metamaterial matches with the design and the measured SEM data.

185

Table 3. Overview of metamaterial structural parameters fitted by scatterometry.

Model	Thickness (nm)	Pillar Width (nm)
SEM	420	45nm
Metamaterial ellipsometry	420-460	NA
Scatterometry (planar layer)	411	NA
Scatterometry (metamaterial fit)	408 - 423	74

186 3.1.3. Rigorous Coupled Wave Analysis

187 RCWA models were created and their structural parameters were fit to averaged Mueller matrix
 188 (MM) ellipsometry measurements (Woollam RC2 - averaged from 30 measurements at 75°
 189 incidence). The ellipsometry data from Figure 2 was used as input during the fits. A fitting
 190 algorithm was created, based on pySCATMECH for calculating Mueller matrices and a robust
 191 Least Square (LSQ) fitting function for optimizing the model. [37, 38] The LSQ functions $F(\mathbf{x})$
 192 is defined as:

$$F(\mathbf{x}) = \sum_{j=1}^m \frac{(y_j - f_j(\mathbf{x}))^2}{\sigma_j^2} \quad (1)$$

193 Here, \mathbf{x} is a vector representing the RCWA structural parameters, y_j represents the measured
 194 MM datapoints at the j -th wavelength, $f_j(\mathbf{x})$ are the calculated datapoints by the RCWA model
 195 and σ_j^2 is the variance of 30 separate measurements. Figure 3 a) shows a schematic of the
 196 RCWA models that were fitted to the measured MM data. Two fits were made with: (1) a model
 197 based on a single SiN layer and (2) a model that assumes the metamaterial has 45nm sized
 198 embedded electrodes in a SiN matrix having 90nm pitch. The pillar width and pitch were not
 199 fitted since the model has only a negligible sensitivity to them. (See Supplemental Document)
 200 The fits and measured MM data are shown in Figure 3 b). It should be noted that MM data is
 201 usually normalized to the first element (MM11) and that there are eight zero elements due to the
 202 symmetry of the measured metamaterial structure. [39]

203

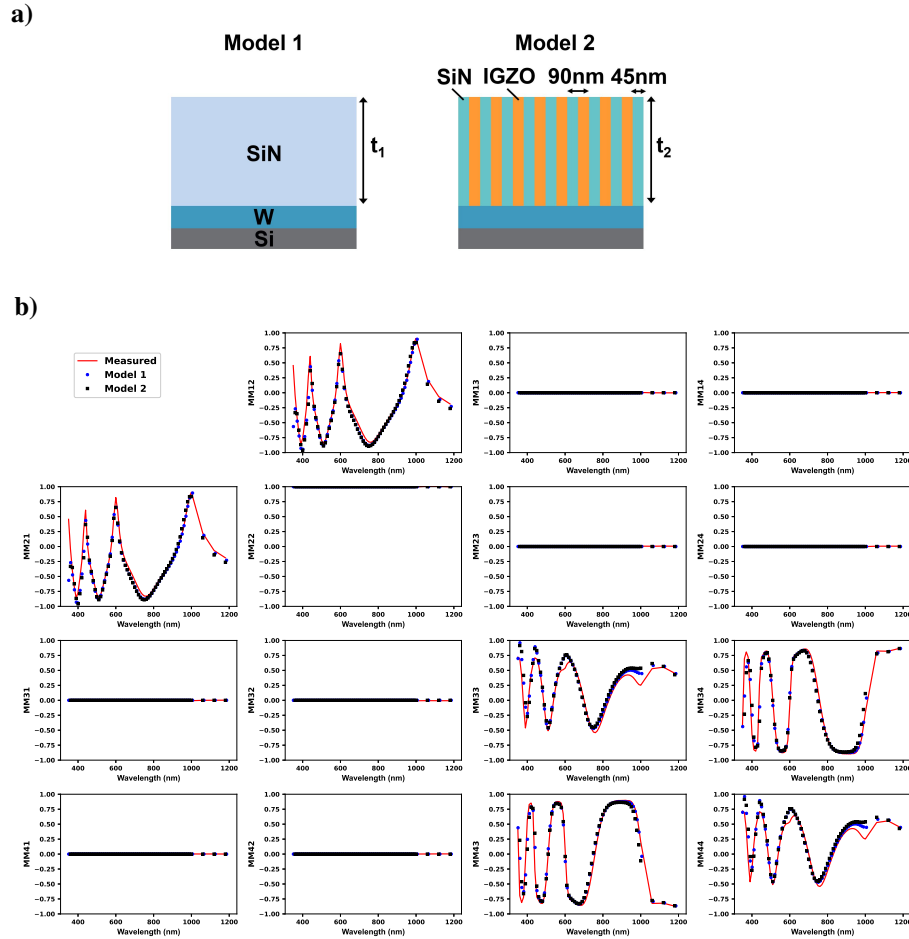


Fig. 3. a) Schematic of the RCWA model assuming the metamaterial is an effective medium. b) Schematic of the RCWA model having IGZO pillars embedded in SiN. c) Mueller matrix measurement and fits of the various models shown in a)-b).

204 For model 1, a good fit with an LSQ value of 4132 was achieved resulting in a thickness of
 205 422nm, proving accurate calculation of the MM data and matching well with our SEM data. For
 206 model 2, which matches the designed ideal metamaterial (45nm electrodes with 90nm pitch),
 207 an LSQ value of 5797 was found at a thickness of 424nm compared to model 1. Clearly, both
 208 models fit the thickness of the metamaterial well. (See Table 4) The closely matching LSQ
 209 values confirm the isotropic nature of the metamaterial. See Supplemental Document for the
 210 LSQ values for various fitted metamaterial heights.

211

212 3.2. Electrical Characterization

213 To prove the conductivity of the metamaterial, conductive atomic force microscopy (C-AFM)
 214 measurements were performed. Figure 4 shows a comparison between an atomic force microscopy
 215 (AFM) and C-AFM measurement. AFM data confirms that the electrode pillar slightly (~ 11 nm)
 216 extrudes from the metamaterial after planarization which aligns well with our SEM data. Figure
 217 4 b) shows a schematic of the electric loop connecting the back contact, pillar electrodes and

Table 4. Overview of metamaterial structural parameters fitted by RCWA simulations.

Model	Thickness (nm)	Pillar Width (nm)
SEM	420	45nm
Metamaterial ellipsometry	420-460	NA
RCWA (planar fit)	422	NA
RCWA (metamaterial fit)	424	No sensitivity

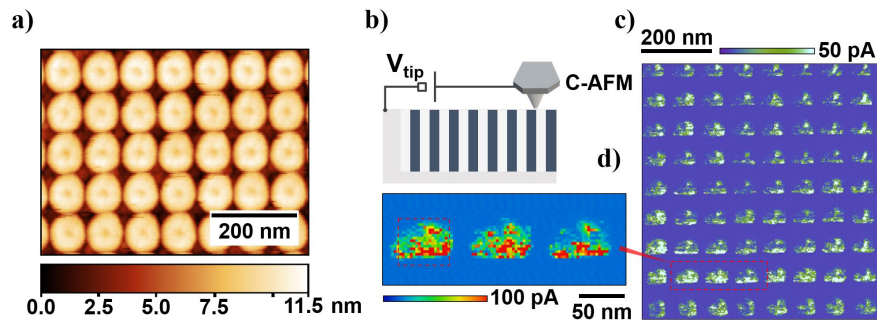


Fig. 4. a) Relative height image of metamaterial by atomic force microscope. b) Schematic of conductive atomic force microscope operation. c) Current map of individual metamaterial pillars by conductive atomic force microscope. d) High resolution conductive atomic force microscope image of the metamaterial.

218 the C-AFM. Here, a nanosized conductive probe is scanned in direct contact with the sample
 219 surface while a voltage is applied between the tip and the sample. Note, that although the
 220 probe is scanning the surface in direct contact, a relatively high tip-sample contact resistance
 221 exists between the tip and the IGZO resulting in low detectable leakage (i.e., in the pA range).
 222 Furthermore, the observed conduction is linked to a convolution of all the present resistive terms.
 223 Hence, this is a combination of the electrode pillar, the metallic back contact and the Si substrate.
 224 Consequently, the dominant resistance is not the tip-sample junction and the absolute value of
 225 current measured should only be considered in relative terms. Additionally, local fluctuations
 226 must be ascribed to small surface modifications of the IGZO top surface such as stoichiometry
 227 variations or intra-grains scattering that can locally make the resistance of the tip-sample
 228 junction dominant. Figure 4 c) shows current flowing at the tip-sample junction when scanning with 6V
 229 bias over an area of $0.8 \times 0.8 \mu\text{m}^2$. Clear contrast is visible corresponding to IGZO, indicating a
 230 lower resistance path for current in these locations and thus confirming the electrode conductivity.
 231 The observed non-round pillar shape is likely linked to the larger than normal employed C-AFM
 232 tip pressure in an attempt to minimize the tip sample resistive junction. Consequently, the slightly
 233 triangular shape is clearly the result of a truncated probe scanned at high pressure, thus losing the
 234 high aspect ratio of a pristine conductive probe. Higher resolution C-AFM imaging is reported
 235 in Figure 4 d) where three IGZO pillars are sensed. The results indicate that the electrical
 236 properties of the IGZO pillars are not uniform, with clear variations of the measured leakage with
 237 fluctuations in the range of 10-20 nm. These are attributed to local structural, and compositional
 238 variations.

239 4. Conclusion

240 We have presented a metamaterial electrode cladding layer designed for waveguide based optical
241 modulators. Three criteria were identified that make for an excellent electrode cladding. Hence,
242 no metals were used close to the active region, a subwavelength electrode pitch was used and
243 Bragg scattering was avoided by refractive index matching the constituent materials, namely
244 SiN and IGZO. We have shown ellipsometry measurements and fits of the metamaterial and
245 its constituents and conclude that it behaves optically isotropic over the visible range. We find
246 that standard ellipsometry oscillators result in a poor fit on the metamaterial, evident from its
247 lower than expected imaginary refractive index. This problem exacerbates when the thickness
248 is included as fitting parameter resulting in a thicker layer with a lowered real refractive index.
249 On the other hand, scatterometry and MM RCWA measurements confirm the dimensions of the
250 embedded pillars and the metamaterial respectively. AFM indicates that the pillar electrodes
251 extrude slightly from its SiN matrix and C-AFM confirms the conductivity of the pillar electrodes.
252 A similar fabrication scheme can be used with alternative materials expanding the capabilities of
253 the metamaterial into different operating regimes, for example ITO can lead to faster switching
254 speeds due to its higher carrier concentration compared to IGZO.

255 5. Backmatter

256 **Funding.** This work has received funding from the European Research Council (ERC) under the European
257 Union's Horizon 2020 research and innovation programme (grant agreement No 742299). G.C. has received
258 an SB PhD fellowship of the FWO (application number 1S91520N).

259 **Disclosures.** The authors have nothing to disclose.

260 **Data availability.** Data underlying the results presented in this paper are available in Dataset 1, Ref. [3].

261 **Supplemental document.** See Supplement 1 for supporting content.

262 References

- 263 1. Q. Jiang, G. Jin, and L. Cao, "When metasurface meets hologram: principle and advances," *Adv. Opt. Photonics* **11**,
264 518 (2019).
- 265 2. J. Sun, E. Timurdogan, A. Yaacobi, E. S. Hosseini, and M. R. Watts, "Large-scale nanophotonic phased array,"
266 *Nature* **493**, 195–199 (2013).
- 267 3. Z. Ren, Y. Chang, Y. Ma, K. Shih, B. Dong, and C. Lee, "Leveraging of MEMS Technologies for Optical Metamaterials
268 Applications," (2020).
- 269 4. W. M. Zhu, A. Q. Liu, X. M. Zhang, D. P. Tsai, T. Bourouina, J. H. Teng, X. H. Zhang, H. C. Guo, H. Tanoto, T. Mei,
270 G. Q. Lo, and D. L. Kwong, "Switchable Magnetic Metamaterials Using Micromachining Processes," *Adv. Mater.*
271 **23**, 1792–1796 (2011).
- 272 5. S. Savo, D. Shrekenhamer, W. J. Padilla, S. Savo, D. Shrekenhamer, and W. Padilla, "Liquid Crystal Metamaterial
273 Absorber Spatial Light Modulator for THz Applications," *Adv. Opt. Mater.* (2014).
- 274 6. Z. He, F. Gou, R. Chen, K. Yin, T. Zhan, and S.-T. Wu, "Liquid Crystal Beam Steering Devices: Principles, Recent
275 Advances, and Future Developments," *Crystals* **9**, 292 (2019).
- 276 7. F. Ding, Y. Yang, and S. I. Bozhevolnyi, "Dynamic Metasurfaces Using Phase-Change Chalcogenides," *Adv. Opt.*
277 *Mater.* **7** (2019).
- 278 8. S. Y. Lee, Y. H. Kim, S. M. Cho, G. H. Kim, T. Y. Kim, H. Ryu, H. N. Kim, H. B. Kang, C. Y. Hwang, and C. S.
279 Hwang, "Holographic image generation with a thin-film resonance caused by chalcogenide phase-change material,"
280 *Sci. Reports* **7** (2017).
- 281 9. J. Hu, S. Bandyopadhyay, Y. H. Liu, and L. Y. Shao, "A Review on Metasurface: From Principle to Smart Metadevices,"
282 *Front. Phys.* **8**, 502 (2021).
- 283 10. X. Zhao, Z. Sun, L. Zhang, Z. Wang, R. Xie, J. Zhao, R. You, and Z. You, "Review on Metasurfaces: An Alternative
284 Approach to Advanced Devices and Instruments," *Adv. Devices & Instrumentation* **2022**, 1–19 (2022).
- 285 11. C. Cho, T. Antrack, M. Kroll, Q. An, T. R. Bärschneider, A. Fischer, S. Meister, Y. Vaynzof, and K. Leo, "Electrical
286 Pumping of Perovskite Diodes: Toward Stimulated Emission," *Adv. Sci.* **8** (2021).
- 287 12. M. Ferrera, S. Saha, A. Boltasseva, V. M. Shalaev, and W. Jaffray, "Transparent conducting oxides: from all-dielectric
288 plasmonics to a new paradigm in integrated photonics," *Adv. Opt. Photonics*, Vol. 14, Issue 2, pp. 148–208 **14**,
289 148–208 (2022).

- 290 13. V. E. Babicheva, A. Boltasseva, and A. V. Lavrinenko, “Transparent conducting oxides for electro-optical plasmonic
291 modulators,” *Nanophotonics* **4**, 165–185 (2015).
- 292 14. M. Dasog, “Transition Metal Nitrides Are Heating Up the Field of Plasmonics,” *Chem. Mater.* **34**, 4249–4258 (2022).
- 293 15. M. Gioti, J. Arvanitidis, D. Christofilos, K. Chaudhuri, T. Zorba, G. Abadias, D. Gall, V. M. Shalaev, A. Boltasseva,
294 and P. Patsalas, “Plasmonic and phononic properties of epitaxial conductive transition metal nitrides,” *J. Opt.* **22**,
295 084001 (2020).
- 296 16. G. V. Naik, V. M. Shalaev, and A. Boltasseva, “Alternative Plasmonic Materials: Beyond Gold and Silver,” *Adv.*
297 *Mater.* **25**, 3264–3294 (2013).
- 298 17. E. Feigenbaum, K. Diest, and H. A. Atwater, “Unity-Order Index Change in Transparent Conducting Oxides at
299 Visible Frequencies,” *Nano Lett.* **10**, 2111–2116 (2010).
- 300 18. G. V. Naik, J. Kim, and A. Boltasseva, “Oxides and nitrides as alternative plasmonic materials in the optical range
301 [Invited],” *Opt. Mater. Express* **1**, 1090 (2011).
- 302 19. Y.-W. Huang, H. W. H. Lee, R. Sokhoyan, R. A. Pala, K. Thyagarajan, S. Han, D. P. Tsai, and H. A. Atwater,
303 “Gate-Tunable Conducting Oxide Metasurfaces,” *Nano Lett.* **16**, 5319–5325 (2016).
- 304 20. J. Kim, S. Choudhury, C. DeVault, Y. Zhao, A. V. Kildishev, V. M. Shalaev, A. Alù, and A. Boltasseva, “Controlling
305 the Polarization State of Light with Plasmonic Metal Oxide Metasurface,” *ACS Nano* **10**, 9326–9333 (2016).
- 306 21. J. Park, J.-H. Kang, S. J. Kim, X. Liu, and M. L. Brongersma, “Dynamic Reflection Phase and Polarization Control
307 in Metasurfaces,” *Nano Lett.* **17**, 407–413 (2017).
- 308 22. M. Y. Shalaginov, S. D. Campbell, S. An, Y. Zhang, C. Ríos, E. B. Whiting, Y. Wu, L. Kang, B. Zheng, C. Fowler,
309 H. Zhang, D. H. Werner, J. Hu, and T. Gu, “Design for quality: Reconfigurable flat optics based on active metasurfaces,”
310 *Nanophotonics* **9**, 3505–3534 (2020).
- 311 23. R. W. Ziolkowski, “Propagation in and scattering from a matched metamaterial having a zero index of refraction,”
312 *Phys. Rev. E* **70**, 046608 (2004).
- 313 24. I. Liberal and N. Engheta, “Near-zero refractive index photonics,” *Nat. Photonics* **11**, 149–158 (2017).
- 314 25. X. Niu, X. Hu, S. Chu, and Q. Gong, “Epsilon-Near-Zero Photonics: A New Platform for Integrated Devices,” *Adv.*
315 *Opt. Mater.* **6**, 1701292 (2018).
- 316 26. H. W. Lee, G. Papadakis, S. P. Burgos, K. Chander, A. Kriesch, R. Pala, U. Peschel, and H. A. Atwater, “Nanoscale
317 conducting oxide PlasMOSor,” *Nano Lett.* **14**, 6463–6468 (2014).
- 318 27. I. C. Reines, M. G. Wood, T. S. Luk, D. K. Serkland, and S. Campione, “Compact epsilon-near-zero silicon photonic
319 phase modulators,” *Opt. Express* **26**, 21594 (2018).
- 320 28. D. C. Zografopoulos, G. Sinatkas, E. Lotfi, L. A. Shahada, M. A. Swillam, E. E. Kriezis, and R. Beccherelli,
321 “Amplitude modulation in infrared metamaterial absorbers based on electro-optically tunable conducting oxides,”
322 *Appl. Phys. A: Mater. Sci. Process.* **124** (2018).
- 323 29. E. H. Turner, “High-Frequency Electro-Optic Coefficients of Lithium Niobate,” *Appl. Phys. Lett.* **8**, 303–304 (1966).
- 324 30. C. Wang, M. Zhang, B. Stern, M. Lipson, and M. Lončar, “Nanophotonic lithium niobate electro-optic modulators,”
325 *Opt. Express* **26**, 1547 (2018).
- 326 31. S. Abel, T. Stöferle, C. Marchiori, C. Rossel, M. D. Rossell, R. Erni, D. Caimi, M. Sousa, A. Chelnokov, B. J. Offrein,
327 and J. Fompeyrine, “A strong electro-optically active lead-free ferroelectric integrated on silicon,” *Nat. Commun.* **4**,
328 1671 (2013).
- 329 32. S. Steudel, J. L. P. Van Der Steen, M. Nag, T. H. Ke, S. Smout, T. Bei, K. van Dienen, G. de Haas, J. Maas, J. de Riet,
330 M. Rovers, R. Verbeek, Y. Y. Huang, S. C. Chiang, M. Ameys, F. de Roose, W. Dehaene, J. Genoe, P. Heremans,
331 G. Gelinck, and A. J. Kronemeijer, “Power saving through state retention in IGZO-TFT AMOLED displays for
332 wearable applications,” *Dig. Tech. Pap. - SID Int. Symp.* **48**, 38–41 (2017).
- 333 33. A. V. Glushkova, H. F. W. Dekkers, M. Nag, J. I. del Agua Borniquel, J. Ramalingam, J. Genoe, P. Heremans,
334 and C. Rolin, “Systematic Study on the Amorphous, C-Axis-Aligned Crystalline, and Protocrystalline Phases in
335 In–Ga–Zn Oxide Thin-Film Transistors,” *ACS Appl. Electron. Mater.* **3**, 1268–1278 (2021).
- 336 34. A. de Jamblinne de Meux, A. Bhoolokam, G. Pourtois, J. Genoe, and P. Heremans, “Oxygen vacancies effects in
337 a-IGZO: Formation mechanisms, hysteresis, and negative bias stress effects,” *physica status solidi (a)* **214**, 1600889
338 (2017).
- 339 35. G. E. Jellison and F. A. Modine, “Parameterization of the optical functions of amorphous materials in the interband
340 region,” *Appl. Phys. Lett.* **69**, 371–373 (1996).
- 341 36. T. E. Tiwald, D. W. Thompson, J. A. Woollam, W. Paulson, and R. Hance, “Application of IR variable angle
342 spectroscopic ellipsometry to the determination of free carrier concentration depth profiles,” *Thin Solid Films*
343 **313-314**, 661–666 (1998).
- 344 37. T. A. Germer, “pySCATMECH: A Python interface to the SCATMECH library of scattering codes,” in *Reflection,*
345 *Scattering, and Diffraction from Surfaces VII*, vol. 11485 L. M. Hanssen, ed. (SPIE, 2020), p. 17.
- 346 38. J. Zhu, S. Liu, X. Chen, C. Zhang, and H. Jiang, “Robust solution to the inverse problem in optical scatterometry,”
347 *Opt. Express* **22**, 22031 (2014).
- 348 39. O. Arteaga, “Useful Mueller matrix symmetries for ellipsometry,” *Thin Solid Films* **571**, 584–588 (2014).